

Development of a rheological model for polymeric fluids based on FENE model

Hossein Ebrahimi¹, Ahmad Ramazani S.A.^{1*}, Seyed Mohammad Davachi^{2,3}

¹Polymer Group, Department of Chemical and Petroleum Engineering, Sharif University of Technology, Tehran, Iran ²Soft Tissue Engineering Research Center, Tissue Engineering and Regenerative Medicine Institute, Central Tehran Branch, Islamic Azad University, Tehran, Iran

³Department of Chemical and Polymer Engineering, Faculty of Engineering, Central Tehran Branch, Islamic Azad University, Tehran, Iran

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ABSTRACT

R heological models for polymer solutions and melts based on the finitely extensible non-linear elastic (FENE) dumbbell theory are reviewed in this study. The FENE-P model that is a well-known Peterlin approximation of the FENE model, indicates noticeable deviation from original FENE predictions and also experimental results, especially in the transient flow. In addition, both FENE and FENE-P models have some shortcomings from the point of view of theory. To overcome these shortcomings, a new approximation of the FENE spring force has been established. It has been used to develop a modified constitutive rheological model for polymeric fluids. In the procedure of modeling, the effect of non-affine deformation is introduced into the new model. Comparison between the model predictions and experimental data presented in the literature for transient and steady shear flow of polystyrene indicates that this modified model can predict the rheological behavior of polymeric fluids with a great accuracy. The newly developed modified model could predict different slopes that can cover the behavior of most of the polymeric fluids. **Polyolefins J (2019) 6: 95-106**

Keywords: Rheological modeling, spring force approximation, FENE model, FENE-P model, FENE-M2 model, FENE modification.

INTRODUCTION

Finitely extensible non-linear elastic dumbbell is a molecular concept of a polymer chain that has been used to describe non-Newtonian rheological effects in polymer solutions and melts [1–7]. FENE model has a vital role in explaining complicated and more realistic physical phenomena of molecular behaviors [8–10]. FENE model predicts the shear-thinning viscosity, first normal stress coefficient and a plateau in the extensional viscosity at high extension rates [11, 12]. This model is reformed to the developed models or systems, such as

FENE-P [13], FENE-L, FENE-LS [14, 15], FENE-S, FENE-D, FENE-QE, FENE-QE- PLA [16, 17], FENE-PM [18], FENE-M, FENE-MR, FENE-LSM, and FENE-LSMR [19]. A complete review of the FENE and its modifications has been done by Venkataramani et al. [3] and Hyon et al. [16]. Except for some simple flows, the FENE model is very challenging and time-consuming to handle and a closure approximation such as the FENE-P model especially when dealing with complex flows is needed [20, 21]. The FENE-P model indicates a noticeable deviation from the original FENE predictions and also experimental results especially in

^{*} Corresponding Author - E-mail: Ramazani@sharif.edu.ir

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the transient flow [14, 15, 19, 20, 22, 23]. In addition, the original FENE and existing closure models have some shortcomings from theoretical aspects.

In this work at first, classical rheological modeling based on FENE theory is reviewed, and some shortcomings of FENE and FENE-P models are investigated. Then, a new approximation of FENE spring force and a rheological constitutive model based on the new approximation are developed. The model is solved in transient and steady shear flows. The effects of the model parameters on the model predictions are investigated. Finally, experimental data that are taken from the literature are compared with model predictions.

THEORETICAL

Review of classical rheological modeling based on FENE theory

The simplest mechanical models for flexible polymer molecules are the elastic dumbbell models, where two beads, each of mass m, are joined by a nonbendable spring [24]. The tension in the spring is denoted by $\mathbf{F}^{\mathbf{C}}$. The location of the beads is given by the position vector r_{i} (v=1, 2), with respect to an arbitrary origin of coordinates and the velocity of the bead is $\dot{\mathbf{r}} = d\mathbf{r} / dt$ of the same origin. The mass center r_c of macromolecule is $r_c = 1/2(r_1 + r_2)$, and its velocity is $\dot{\mathbf{r}}_c$. The connector vector or end-to-end vector $Q=r_2-r_1$ describes the overall orientation and internal configuration of the polymer molecules. Each bead is presumed to experience a Stokes' law drag force. According to Stokes' law the drag force is assumed to be proportional to the relative velocity between solvent and bead, with a proportionality constant ζ called the friction coefficient [24, 25]. The beads represent molecular segments having several monomers, and the spring describes the entropic effects in which the end-to-end vector of the polymer is the subject [22]. The flow field of the polymer solution is taken to be homogeneous, then the mass average velocity of the solution can be written as $v = v_0 + [\kappa.r]$ where v_0 is a vector independent of position and $\kappa^t = \nabla v$ is the velocity gradient tensor which is independent of the position vector r in the fluid, and its trace has zero value due to the assumption of fluid incompressibility. The Fokker-Planck equation governing the evolution of the configurational distribution function $\Psi(\mathbf{Q},t)$ is [24, 26]:

$$\frac{\partial \psi}{\partial t} = -\left(\frac{\partial}{\partial Q} \cdot \left\{ [\mathbf{\kappa} \mathbf{Q}] \psi - \frac{2k_B T}{\zeta} \frac{\partial}{\partial Q} \psi - \frac{2}{\zeta} \mathbf{F}^c \psi \right\}$$
(1)

Where k_B is the Boltzmann constant, and T is the temperature in Kelvin. In equation (1) the first term on the right-hand side accounts for the hydrodynamic drag forces on the beads due to the dumbbell's movement through the solvent. The second term describes the Brownian motion force on the beads because of thermal fluctuations in the solvent, and the last one accounts for the forces transmitted through the connecting spring. The diffusion coefficient $k_BT/2\zeta$ is assumed to be constant. For a function B(Q) of the connector vector, the time-dependent configuration space average is given by the following equation:

$$\langle B \rangle = \int B(\mathbf{Q})\psi(\mathbf{Q},t)d^{3}\mathbf{Q}$$
 (2)

Therefore, from the Fokker-Planck equation [24, 26], the equation of motion for the second moments of the distribution function for the end-to-end vector can be obtained as follows:

$$\frac{d}{dt} \langle \mathbf{Q} \mathbf{Q} \rangle = \kappa \cdot \langle \mathbf{Q} \mathbf{Q} \rangle + \langle \mathbf{Q} \mathbf{Q} \rangle \cdot \kappa^{t} + \frac{4k_{B}T}{\zeta} \delta - \frac{4}{\zeta} \langle \mathbf{Q} \mathbf{F}^{c} \rangle \quad (3)$$

Where $\langle \mathbf{QQ} \rangle$ is the second moment of configuration tensor and δ is the unit tensor. The polymer contribution to the stress tensor, from the Kramer's expression, can be obtained as [24]:

$$\sigma_p = -n \langle \mathbf{Q} \mathbf{F}^C \rangle + nk_B T \delta \tag{4}$$

Where n is the number of macromolecules in the unit volume. To complete the model presentation, it is enough to introduce the spring force into equations (3) and (4). The simplest form of spring force is linear Hookean, that denotes by $F^{C(Hookean)}$ and takes the following manner [24, 27]:

$$\mathbf{F}^{\mathrm{C}(\mathrm{Hookean})} = H\boldsymbol{Q} \tag{5}$$

Where H is the spring constant. On the contrary with polymeric fluid behavior, the Hookean dumbbell model predicts constant values for all of the viscometric functions. This model can be only realistic for small deformations from the equilibrium (Gaussian distribution) and puts no limit to the extent in which the dumbbell can be stretched [11, 28]. To describe non-Newtonian rheological effects in polymer solutions, the finitely extensible non-linear elastic (FENE) spring force was introduced by Warner et al. [29]:

$$\mathbf{F}^{\mathrm{C(FENE)}} = \frac{H}{1 - Q^2 / Q_0^2} \boldsymbol{Q}$$
(6)

Where Q_0 is the maximum extension of polymer chains. By using the spring force equation (6), there is no closed constitutive equation for the polymeric stress tensor, and no simple analytical solutions are possible [24]. An analytically more tractable dumbbell model which leads to a final constitutive equation can be attained by substituting the configuration-dependent non-linear factor in the FENE spring force by a selfconsistently averaged term [24]:

$$\mathbf{F}^{\mathrm{C}(\mathrm{FENE-P})} = \frac{H}{1 - \langle Q^2 / Q_0^2 \rangle} \mathbf{Q}$$
(7)

This pre-averaging spring force is known as the Peterlin approximation and the resulting model as the FENE-P model. In the FENE and FENE-P models, $\lambda_{\rm H} = \zeta/4$ H and b=HQ₀²/K_BT called relaxation time and extensibility parameter, respectively. Although most widely used non-linear kinetic models are FENE and its closure approximations, there are some shortcomings in the FENE and FENE-P models: i. Based on Larson [30], when a chain end extends to increase the end-to-end distance Q, the needed force to overcome the entropic "spring" force of the chain

$$\mathbf{F}^{C} = 2K_{B}T\beta^{2}\mathbf{Q}$$
(8)

where,

is as follows:

$$\beta^2 = \frac{3}{2Q^2} \tag{9}$$

By replacing the β from equation 9 to 8, the entropic force will be obtained as follows:

$$\mathbf{F}^{\mathrm{C}} = \frac{3\mathrm{K}_{\mathrm{B}}\mathrm{T}}{\mathrm{Q}^{2}}\mathbf{Q} \tag{10}$$

On the other hand, at equilibrium (steady state with $\kappa=0$), $\frac{d}{dt}\langle QQ \rangle = 0$, therefore, based on equation 3, the can be easily obtained as:

$$\left\langle \mathbf{Q}\mathbf{F}^{C}\right\rangle _{eq}=k_{B}T\delta \tag{11}$$

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It is known that, at equilibrium, $Q^2 = tr \langle QQ \rangle_{eq}$, accordingly, Q^2 will be generated as follows:

$$Q^2 = \frac{3K_B T}{H} \tag{12}$$

Finally, from equations 10 - 12:

$$\langle \mathbf{Q}\mathbf{Q} \rangle_{eq} = (k_B T / H)\delta$$
 (13)

According to Herrchen and Öttinger (Herrchen and Öttinger 1997), the equilibrium state for the second moment of the FENE is $\langle \mathbf{QQ} \rangle_{eq} = (k_B T / H)[b/(b+5)]\delta$ and those of for the FENE-P model is $\langle \mathbf{QQ} \rangle_{eq} = (k_B T / H)[b/(b+3)]\delta$. Therefore, the equation 13 for the FENE and FENE-P model will be satisfied just for $b \rightarrow \infty$. The obtained results are in accordance with the previous reports by Stephanou et al. [31]. In the real condition, the spring can be pulled to fully extent state, this means the polymer chain is in the fully stretched-out conformation. In this condition, FENE and FENE-P models calculate infinite value for spring force, in contrary with real condition.

ii. In the FENE and FENE-P models, the maximum extension of the polymer chain is Q_0 , then at $Q^2 = Q_0^2$ the spring force will be infinite based on mathematical equations because Fc will be infinite, but indeed, the spring force should have a finite value in the physical or real conditions.

iii. The FENE-P model can be considered as a good approximation to the original FENE dumbbell model only in steady sate flows, while larger deviation can be seen in time-dependent flows [14, 15, 19, 22]. Van Heel et al. studied the start-up of shear and elongational flow of FENE and FENE-P by Brownian dynamics simulations and revealed that the FENE-P response could be so different from the FENE behavior [32]. They indicated that at the inception of the flow, the overshoot and the time which overshoot occurs, were more than those of FENE for the shear stresses. Also, the constant values of FENE-P predictions were higher than those of FENE. Herrchen and Öttinger indicated that the prediction of the FENE and FENE-P for material functions after the inception of flow is no longer quantitatively close; the time-dependent viscometric functions in the start-up of shear flow at high shear rates are over predicted by more than a factor of two by the FENE-P approximation. So, Peterlin approximation is not an appropriate closure for the FENE model [22].



iv. The original FENE and FENE-P models predict only a constant value for the slope of the linear or power-law region in steady shear flow. In addition, the prediction of the models for second normal stress coefficient ψ_2 is zero, and then they could not predict any value for ψ_2/ψ_1 .

To overcome the above shortcomings of the FENE and FENE-P models, a more convenient approximation of FENE spring force has been developed. Then a rheological model based on modified spring force has been presented. In addition, the effects of the non-affine deformation have been introduced into the modified model.

MODEL FORMATION

Spring force

As mentioned before, to overcome shortcomings of the FENE and FENE-P models, a new spring force approximation was introduced, which henceforth denotes $F^{C(FENE-M2)}$, as follows:

$$\mathbf{F}^{C(FENE-M2)} = \frac{H}{1 - \frac{\langle Q^2 \rangle - \langle Q^2 \rangle eq}{Q_0^2 + \varepsilon \langle Q_{eq}^2 \rangle}} \mathbf{Q}$$
(14)

Where ε is a phenomenological parameter. On the contrary with the FENE and FENE-P models, in this modified introduced spring force the second moment of the end-to-end vector of polymer chain at equilibrium condition is $\langle \mathbf{Q}\mathbf{Q} \rangle_{eq} = (\mathbf{k}_{\rm B} \mathrm{T}/\mathrm{H})\delta$. In addition, the spring force would have a finite value at $Q^2 = Q_0^2$. In the new form of spring force presentation, it has been assumed that the polymer chain can be extended a few more than Q_0 . At $Q^2 = Q_0^2$ the polymer chain is under the fully stretched-out conformation. If the stretching continues, the bonds of the chain will start to extend. The maximum value of the chain extension is indicated by $Q_{\rm max}^2 = Q_0^2 + \varepsilon \langle Q_{\rm eq}^2 \rangle$ or $Q_{\rm max}^2 = b + 3\varepsilon$ in dimensionless form. Therefore, the acceptable values for ε are ε >-b/3.

Constitutive equation presentation

The following dimensionless forms of variables and parameters have been introduced to simplify the model presentation:

$$\mathbf{Q}^* = \frac{Q}{(K_B T / H)^{1/2}}; t^* = \frac{t}{\lambda_H}; \sigma_p^* = \frac{\sigma_p}{nk_B T}; \mathbf{\kappa}^* = \lambda_H k \quad (15)$$

Replacing equation (14) in (3) results in:

$$\frac{d}{dt^*} \langle \mathbf{Q}^* \mathbf{Q}^* \rangle = \mathbf{\kappa}^* \langle \mathbf{Q}^* \mathbf{Q}^* \rangle + \langle \mathbf{Q}^* \mathbf{Q}^* \rangle \mathbf{\kappa}^{*\prime} + \delta - \frac{b + 3\varepsilon}{b + 3\varepsilon + 3 - tr} \langle \mathbf{Q}^* \mathbf{Q}^* \rangle \langle \mathbf{Q}^* \mathbf{Q}^* \rangle$$
(16)

The contribution of polymer in the stress tensor can be demonstrated as follows in case of replacing equation (14) in (4):

$$\sigma_{p}^{*} - \frac{b + 3\varepsilon}{b + 3\varepsilon + 3 - tr \langle \mathbf{Q}^{*} \mathbf{Q}^{*} \rangle} \langle \mathbf{Q}^{*} \mathbf{Q}^{*} \rangle + \delta$$
(17)

Where tr represents trace. To simplify more in equations (18) and (19), $\mathbf{M}^* = \langle \boldsymbol{\varrho}^* \boldsymbol{\varrho}^* \rangle$ is assigned. Also, the transpose of the velocity gradient tensor k^{*} can be written as follows:

$$\boldsymbol{\kappa}^* = 1/2\lambda_H(\dot{\boldsymbol{\gamma}} - \boldsymbol{\omega}) \tag{18}$$

Where $\dot{\gamma}$ is the rate of strain tensor and is the vorticity tensor. Now equations (16) and (17) are rewritten in a new form as follows:

$$\frac{d\mathbf{M}^*}{dt^*} = \frac{1}{2}(\dot{\gamma}.\mathbf{M}^* + \mathbf{M}^*.\dot{\gamma}) - \frac{1}{2}(\omega.\mathbf{M}^* - \mathbf{M}^*.\omega) + \delta - \frac{b+3\varepsilon}{b+3\varepsilon+3-t}\mathbf{M}^* \mathbf{M}^*$$
(19)

$$\sigma_p^* = -\frac{b+3\varepsilon}{b+3\varepsilon+3-tr\mathbf{M}^*}\mathbf{M}^* + \delta$$
(20)

The developed model is called FENE-M2 approximation on FENE spring force.

Introducing the non-affine deformation

In the study of dumbbell kinetic theory, it has been assumed that the two beads of the dumbbell move through the solvent without disturbing the velocity field [33, 34]. Therefore, the velocity of fluid in each bead is as same as the velocity of fluid (that is $v=v_0+[\kappa.r]$). An alternative procedure to introduce the effect of the flow perturbation at bead v resulting from the motion of the other beads is to replace the fluid velocity at bead by the following equation:

$$\mathbf{v}_{v} = \mathbf{v}_{0} + [\mathbf{\kappa} \cdot \mathbf{r}_{v}] - \frac{1}{2} \zeta [\dot{\boldsymbol{\gamma}} (\mathbf{r}_{v} - \mathbf{r}_{c})]$$
(21)

Where ζ is a small quantity that called Gordon-Schowalter (GS) parameter [24, 35, 36]. This parameter is also known as the "Slip parameter" which makes the corresponding stress related to the non-affine motion [37]. With no slippage (ζ =0), the motion of the polymer molecules is said to be affine.

If this alternative procedure is introduced to the FENE-M2 model, the time evolution equation (21) will be as follows:

$$\frac{d\mathbf{M}^*}{dt^*} = \frac{1}{2}(\dot{\gamma}.\mathbf{M}^* + \mathbf{M}^*.\dot{\gamma}) - \frac{1}{2}(\omega.\mathbf{M}^* - \mathbf{M}^*.\omega) + \delta - \frac{b + 3\varepsilon}{b + 3\varepsilon + 3 - tr\mathbf{M}^*}\mathbf{M}^*$$
(22)

By this modification, the value of the second normal stress coefficient will be non-zero and $\Psi_2/\Psi_1=-1/2\zeta$ [24, 35, 36]. Constitutive equations, which have the GS derivative, show unreal oscillations in the start-up shear flow material functions curve [38, 39]. This problem will be investigated in Section 4.2.

RESULTS AND DISCUSSION

Equation (22) represents a system of ordinary differential equations that is solved numerically both for transition and steady shear flows. The initial condition is the equilibrium state $M_{eq}^{*}=\delta$ and the model parameters are b, ε and ζ . In simple shear flow, the velocity field is given by $v_x = \dot{\gamma}_{yx}y$, $v_y = 0$ and $v_z = 0$ where $\dot{\gamma}_{yx}$ is the velocity gradient and can be a function of time. The absolute value of $\dot{\gamma}_{yx}$ is called the shear rate and denoted by $\dot{\gamma}$. Thus, the rate of strain tensor and the vorticity tensor are as follows:

$$\dot{\gamma}(t) = \dot{\gamma}(t) \begin{pmatrix} 0 & 1 & 0 \\ 1 & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix}, \\ \omega(t) = \dot{\gamma}(t) \begin{pmatrix} 0 & -1 & 0 \\ 1 & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix}$$
(23)

The stress tensor has the following form with $\sigma_{xy} = \sigma_{yx}$.

$$\sigma = \begin{pmatrix} \sigma_{xx} & \sigma_{xy} & 0\\ \sigma_{yx} & \sigma_{yy} & 0\\ 0 & 0 & 1 \end{pmatrix}$$
(24)

For steady shear flow (sometimes called a viscometric flow) the shear rate does not depend on the time, it is presumed that the shear rate has been constant for so long that all the stresses in the fluid are time-independent [24, 40]. The dimensionless shear viscosity η is defined analogously to the viscosity of Newtonian fluids:

$$\boldsymbol{\eta}^*(\dot{\boldsymbol{\gamma}}) = -\frac{\sigma_{p,xy}^*}{\dot{\boldsymbol{\gamma}}^*} \text{ or } \frac{\eta}{nk_B T \lambda_H} = -\frac{\sigma_{p,xy}}{nk_B T \lambda_H \dot{\boldsymbol{\gamma}}}$$
(25)

Likewise, the dimensionless first and second normal stress coefficients ψ_1^* and ψ_2^* can be defined as follows, Where $\dot{\gamma}^* = \lambda_H \dot{\gamma}$ is the dimensionless shear rate and σ_{ij}^* are the components of the extra stress tensor given by equation (20):

$$\boldsymbol{\psi}_{1}^{*}(\dot{\boldsymbol{\gamma}}) = -\frac{\sigma_{p,xx}^{*} - \sigma_{p,yy}^{*}}{\dot{\boldsymbol{\gamma}}^{*2}} \text{ or } \frac{\psi_{1}(\dot{\boldsymbol{\gamma}})}{nk_{B}T\lambda_{H}^{2}} = -\frac{\sigma_{p,xx}^{*} - \sigma_{p,yy}}{nk_{B}T(\lambda_{H}\dot{\boldsymbol{\gamma}})^{2}}$$
(26)

$$\Psi_2^*(\dot{\mathbf{y}}) = -\frac{\sigma_{p,yy}^* - \sigma_{p,zz}^*}{\dot{\gamma}^{*2}} \text{ or } \frac{\Psi_2(\dot{\gamma})}{nk_B T \lambda_H^2} = -\frac{\sigma_{p,yy} - \sigma_{p,ZZ}}{nk_B T (\lambda_H \dot{\gamma})^2}$$
(27)

For each transient shear flow as in steady shear flow the relevant material functions are the viscosity η^{\pm} , and the first and second normal stress coefficients Ψ_1^{\pm} and Ψ_2^{\pm} . The plus and minus signs are used for start-up and relaxation material functions, respectively. These are given in dimensionless form by equation (28):

$$\boldsymbol{\eta}^{*\pm}(\dot{\boldsymbol{\gamma}}) = -\frac{\sigma_{p,xy}^{*}}{\dot{\gamma}^{*}}, \boldsymbol{\psi}_{1}^{*\pm}(\dot{\boldsymbol{\gamma}}) = -\frac{\sigma_{p,xx}^{*} - \sigma_{p,yy}^{*}}{\dot{\gamma}^{*2}}, \boldsymbol{\psi}_{2}^{*\pm}(\dot{\boldsymbol{\gamma}}) = -\frac{\sigma_{p,yy}^{*} - \sigma_{p,zz}^{*}}{\dot{\gamma}^{*2}}$$
(28)

Therefore, in transient flow the curves of η , Ψ_1 and Ψ_2 versus time describe the time-dependent rheological behavior of polymeric materials. In the following subsections, the effects of the influencing parameters on the model prediction will be investigated.

Effects of the slip parameter on the model predictions

As it is mentioned, introducing ξ into the model causes an oscillation in the material functions curve versus time in the start-up shear flow. Figure 1(a) indicates that at the given shear rate for small values of ξ , there is no oscillation in the curve of shear stress versus time. The oscillations in the curve are appeared as ξ is increased.

Besides, the size of overshoot and steady value of the shear stress are decreased by increasing the value of slip parameter from zero (affine deformation) to up. In fact, the model shows increased shear thinning behavior as the "slippage" of polymer strands in the continuum takes into account correctly [41, 42]. Figure 1(b) demonstrates that at a fixed value of ξ , b and ε the frequency of oscillation is increased via increasing the shear rate. The frequency and amplitude of oscillation are a function of shear rate, b and ε parameters. It has been found by mathematical solution of constitutive equation of the modified model that for given values of b and ε , the value of ξ shows no oscillation and is



Figure 1. Oscillations in the curve of shear stress vs. time in the start-up shear flow with model parameters b=50 and ϵ =0:(a) Dimensionless shear rate has fixed value $\lambda_n \dot{\gamma} = 15$ and slip parameter has different values. (b) Slip parameter has fixed value ξ =0.04 and dimensionless shear rate has different values.

about $\zeta < 15/[(b+\varepsilon)\lambda_H\dot{\gamma}]$ [43–45].

Figures 2(a) and 2(b) depict the influence of slip parameter ξ on the transient material functions, η^{\pm} and Ψ_1^{\pm} , respectively. The results indicate that the size of overshoot and steady state plateau values are decreased via increasing ξ . In Figure 3 the effect of the slip parameter on the shear rate dependent viscosity is presented. As it is seen for selected model parameters, increasing ξ from zero to 0.004 changes the slope of curve from -0.66 to -0.98.

Effects of the ε parameter on the model predictions The role of ε which is a new parameter introduced into the FENE-M2 model, is investigated. To clarify the effect of ε on the model predictions, a comparison between FENE, FENE-P and FENE-M2 model's predictions in transient shear flows is done. As mentioned before, to choose a value for ε , one should notice that the acceptable values for ε are ε >-b/3. Figure



Figure 2. Effect of the slip parameter on the stress growth and relaxation of material functions with the dimensionless shear rate $\lambda_{ii}\dot{\gamma} = 5$ and the model parameters b=50 and ε =0: (a) viscosity and (b) first normal stress coefficient.

4(a) exhibits the evolution of the time in the first normal stress difference during start-up of shear flow for three models. The results indicated the FENE-P model is over prediction than the FENE model, especially at higher shear rates. The more shear rate increases, the more value of over predictions can be seen. It could be seen that the FENE-M2 predictions are so close to those of the FENE model for chosen parameters. It is possible to make the FENE-M2 predictions closer into the FENE or FENE-P by selecting different values of ε . Figure 4(b) shows the FENE-M2 and FENE-P predictions for the first normal stress difference with respect to chosen model parameters which are nearly coincident when .

The effect of ε parameter on the transient material functions, η^{\pm} and Ψ_1^{\pm} , is survived in Figures 5(a) and 5(b), respectively. An increasing in ε value grows the steady value of material functions. Besides, for both material functions, the overshoot occurs at a later time with increasing ε .

Figure 6(a) and (b) represents the effect of ε on



Figure 3. Effect of the slip parameter on the steady viscosity with the model parameters b=50 and e=6.

the steady material functions, η and Ψ_1 , respectively. As it can be seen, increasing of ε would increase the



Figure 4. Comparison between the FENE, FENE-P and FENE-M2 models in the prediction of the first normal stress difference in the start-up of shear flow. The model parameters are b=50 and ξ =0 and the dimensionless shear rates are $\lambda_{H}\dot{\gamma} = 1,2$ and 4 (from down to top): (a) With ϵ =-3.5 the results of FENE-M2 and FENE models are close them (b) With ϵ =-11.2 the results of FENE-M2 and FENE-M2 and FENE-P models nearly are coincident. Data for FENE model reproduced from van Heel et al. [32].

length of the plateau with no change in the slope of the curve. In the other hand, the "power law exponent" is independent of ε values.

Effects of the extensibility parameter on the model predictions

The transient viscosity and the first normal stress difference coefficient at different values of extensibility parameter are presented in Figures 7(a) and 7(b). As it is seen, for both material functions, η^{\pm} and Ψ_{1}^{\pm} , the steady values grow with increasing b. In addition, the overshoot occurs at a later time with increasing b.

Figures 8(a) and 8(b) display the influence of extensibility parameter on the steady viscosity and the first normal stress difference, respectively. Similarly, to the ε effect, for both material functions increasing of b would increase the length of plateau. Also, the exponent of the power law is again independent of the extensibility parameter b.



Figure 5. Effect of ε parameter on the stress growth and relaxation of material functions with dimensionless shear rate $\lambda_H \dot{\gamma}$ =4 and model parameters b=20 and ξ =0: (a) viscosity and (b) first normal stress coefficient.





Figure 6. Effect of e parameter on the steady material functions with model parameters b=20 and ξ =0: (a) viscosity and (b) first normal stress coefficient.

Effects of the shear rate on the model predictions

Figures 9(a) and 9(b) show the effect of the shear rate on the transient material functions, η^{\pm} and Ψ_{1}^{\pm} , respectively.

The results indicate that there is no overshoot or

related phenomenon for low shear rates. The viscosity and the first normal stress difference show an overshoot when the shear rate increases to its larger values. The size of this overshoot increases by increasing the shear rate while the time corresponding to its maximum





Figure 7. Effect of the extensibility parameter b on the material functions with dimensionless shear rate $\lambda_{\rm H}\dot{\gamma}$ =4 and model parameters ε =6 and ξ =0: (a) viscosity and (b) first normal stress coefficient.

Figure 8. Effect of the extensibility parameter b on the steady material functions with model parameters ε =6 and ξ =0: (a) viscosity and (b) first normal stress coefficient.



Figure 9. Effect of the shear rate on the transient material functions. Model parameters are b=50, ε =0 and ξ =0: (a) viscosity and (b) first normal stress coefficient.

value decreases as the shear rate increases [41, 46]. Besides, relaxation curves show that both material functions relax monotonically to zero in low shear rate and then it relaxes more rapidly as the shear rate increases.



Figure 10. Comparison between the model prediction and experimental date in the start-up viscosity shear flow. Model parameters are b=20, ε =4 and ξ =0.03. Data are taken from Schweizer et al. [47].



Figure 11. Comparison between the model prediction and experimental date for the first normal stress difference in start-up shear flow with $\dot{\gamma}$ =10s⁻¹. Model parameters are b=20, ε =4 and ξ =0.03. Data are taken from Schweizer et al. [47].

Comparison with experimental data

The comparison between the model prediction and experimental data (taken from Schweizer et al. [47]) in the start-up shear flow is presented in Figure 10 for polystyrene melt at 175° C and three shear rates 0.1, 1 and 3 s⁻¹. The results indicate that the experimental data are in good agreement with model predictions.

Figure 11 compares the model prediction in start-up shear flow for the first normal stress difference, N_1 , and the experimental data taken from Schweizer et al. [47]. The presented results show that model predictions are in qualitative and quantitative agreement with experimental results.

Figure 12 presents the model predictions with counterpart experimental data taken from Laun study for LDPE melt at 150°C in the steady shear flow for the viscosity and the first normal stress difference [48].



Figure 12. Comparison between the model prediction and experimental data for the viscosity and the first normal stress difference in the steady shear flow. Model parameters are b=18, ε =0 and ξ =0. Data are taken from Schweizer et al. [47].



As it is seen, the model predictions are in excellent agreement with experimental data.

CONCLUSION

In the review of the FENE and FENE-P models, some shortcomings of these models are highlighted. Prediction results in the transient flow indicate a significant difference between model predictions by original FENE and its approximation FENE-P. A new closure approximation for spring force has been developed and used to develop a modified FENE based (FENE-M2) model. The presented results indicate that this closure approximation should be closer to the original FENE than Peterlin approximation. In the procedure of modeling, the effect of non-affine deformation is studied and a parameter called slip parameter is introduced into the model to improve model prediction behavior. Where the FENE and FENE-P models predict only a constant slope for material functions in steady shear flow curves, the newly developed modified model could predict different slopes that can cover the behavior of most polymeric fluids. Comparison of the model predictions with experimental data shows the FENE-M2 model can predict the behavior of polymeric fluids in the steady and transient shear flow correctly.

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